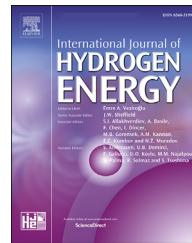




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Partial crystallization of Co–Fe oxyhydroxides towards enhanced oxygen evolution activity

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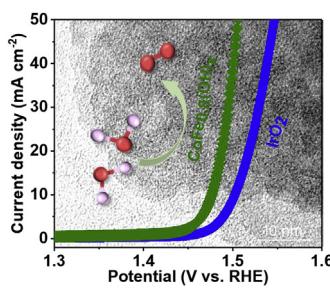
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HIGHLIGHTS

- Partially crystallized $\text{CoFe}_{0.8}(\text{OH})_x$ shows higher OER activity than amorphous one.
- $\text{CoFe}_{0.8}(\text{OH})_x$ exhibits an overpotential of only 246 mV at 10 mA cm^{-2} .
- Bimetallic $\text{CoFe}_{0.8}(\text{OH})_x$ shows lower OER barrier than mono-metallic counterpart.
- The charge transfer ability can be enhanced by partial crystallization.
- The OER active sites should be di- μ -oxo bridged Fe–Co.

GRAPHICAL ABSTRACT



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ABSTRACT

Developing high-performance noble-metal-free electrocatalysts for the oxygen evolution reaction (OER) is of great significance for the large-scale implement of electrochemical water splitting. Here, we demonstrate that cobalt-iron oxyhydroxide ($\text{CoFe}_{0.8}(\text{OH})_x$) with appropriate crystallization exhibits excellent OER activity with a low overpotential of only 246 mV, which is much lower than that achieved on amorphous one. Kinetic experiments indicate that the OER active sites should be di- μ -oxo bridged Fe–Co, on which lower energy barrier is attained than that on Fe–O and Co–O sites. Partial crystallization is beneficial to the lattice contraction, the formation of Co–O–Fe sites, and the decrease of charge transfer resistance, thus accelerating OER process.

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Introduction

Hydrogen is an ideal energy carrier and one of the most promising alternatives for fossil fuels since the combustion product is water [1–6]. The electrolysis of water is a facile and efficient technique to generate highly pure hydrogen by splitting water into hydrogen and oxygen via two half-reactions: oxygen evolution reaction (OER) hydrogen evolution reaction (HER) [7–11]. Due to slow kinetics and involving 4e transfer, OER is regarded as the rate-determining step of water splitting [12,13]. Although RuO₂ and IrO₂ can catalyze OER well in acidic media, they are unstable in alkaline media [14,15]. Moreover, the scarcity and high cost hinder their large-scale industrial applications [16,17]. Therefore, a lot of investigations have been carried out to develop earth-abundant and cost-efficient transition metal-based catalysts to speed up the sluggish OER [18–25].

The cobalt oxide was theoretically predicted to be very active for the OER, however, the experimental overpotential on CoO_x catalysts is size-dependent [12,26]. In principle, the smaller the CoO_x nanoparticles are, the better the OER activity is. In addition, compared with crystalline cobalt oxide, amorphous CoO_x usually showed higher OER activity due to enhanced stabilization of the O* intermediate on the catalyst surface and increased active sites [26]. Although single-metal Co-based (oxy)hydroxides have been widely investigated for many years, their OER activity is still inferior to commercial iridium oxide, which cannot meet the actual application needs. Therefore, bimetallic Co-based mixed (oxy)hydroxides

have received great attention [27]. Various transition metals doped Co-based (oxy)hydroxides were synthesized for the OER, e.g. Cr-doping [28], Fe-doping [29], Ni-doping [30], Mn-doping [31], Cu-doping [32], Sb-doping [33], Mo-doping [34], Ce-doping [35], Ru-doping [36], and Ir-doping [37]. Thereinto, Fe-doping is a very efficient strategy to significantly improve the OER performance of Co-based electrocatalysts due to a cooperation mechanism between Fe and Co during the reaction [29,38].

For Co–Fe (oxy)hydroxides, the OER performance is greatly influenced by the structure [39–41]. For example, Sargent et al. found that a gelled FeCo oxyhydroxide exhibited an OER overpotential (η_{10}) of 277 mV at 10 mA cm⁻² [42]. Zhu et al. reported that ultrathin CoFeO_x nanosheets with thickness of 1.2 nm and abundant oxygen vacancies showed an η_{10} of 350 mV [43]. Zeng et al. found that crystalline CoFe nanoparticles wrapped in N-doped carbon nanotubes exhibited an η_{10} of 292 mV [44]. We found that a core-shell Co₃O₄@FeO_x catalyst displayed higher OER activity than CoFe-mixed oxides, which was ascribed to better ability to bind water molecules on Fe sites and improved charge transfer ability [45]. Furthermore, we found that a mesoporous CoFe_{0.05}O_x demonstrated an η_{10} of 280 mV due to a cooperation mechanism between Fe and Co sites to lower the OER barrier [46]. Compared with crystalline metal oxides, amorphous counterparts showed higher OER activity [47]. However, amorphous mixed metal oxides usually exhibit larger resistance than crystalline ones due to irregular atomic distribution, resulting in poor charge transfer property [48]. Therefore, it is

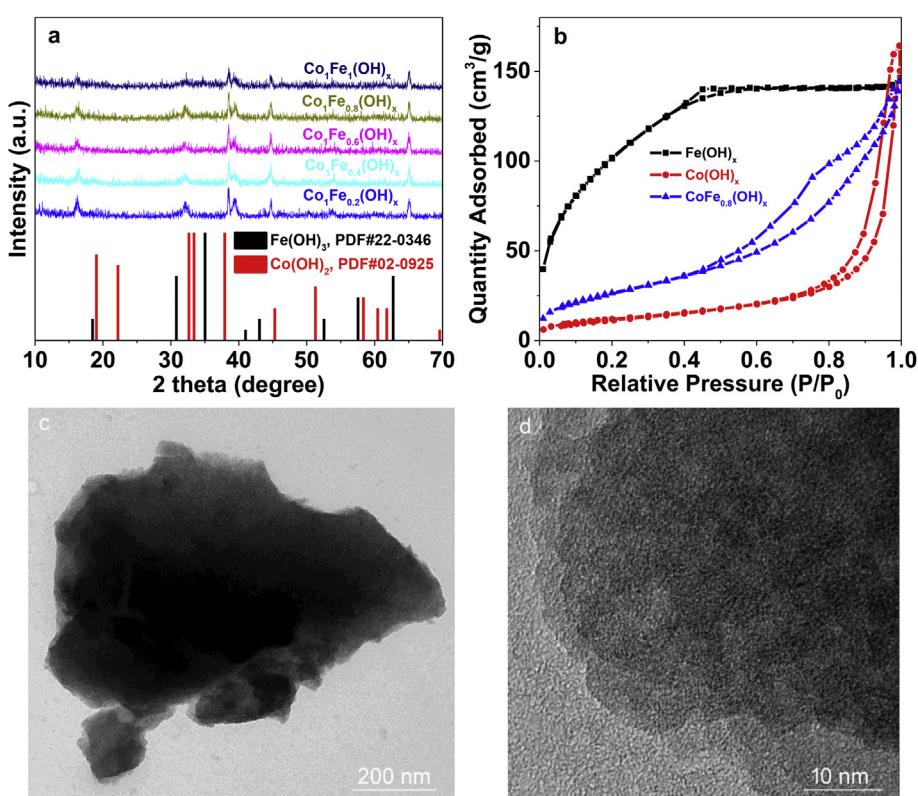


Fig. 1 – XRD patterns of CoFe_n(OH)_x and referred samples. (b) N₂ adsorption–desorption isotherms of Fe(OH)_n, Co(OH)_n, and CoFe_{0.8}(OH)_x. (c) TEM image of CoFe_{0.8}(OH)_x. (d) HRTEM image of CoFe_{0.8}(OH)_x.

advisable to design partially crystallized materials combining the merits of good charge transfer ability in crystalline mixed metal oxides and high OER activity of amorphous mixed metal oxides [49].

Here, we fabricated partially crystallized Co–Fe oxyhydroxides for enhanced OER activity. Compared with amorphous Co–Fe oxyhydroxides, partially crystallized ones show higher OER activity due to faster charge transfer and lower reaction barrier. According to the kinetic studies, single-component Fe/Co oxyhydroxide exhibits much higher OER barrier than Co–Fe oxyhydroxides, indicating that the efficient active site should be Fe–O–Co species.

Results and discussion

Partially crystallized Co–Fe oxyhydroxides were synthesized by drying the gelled Co–Fe at 40 °C, which were nominated as $\text{CoFe}_n(\text{OH})_x$ ($n = 0.2, 0.4, 0.6, 0.8$, and 1), while amorphous $\text{CoFe}_{0.8}(\text{OH})_x$ oxyhydroxide was synthesized by a vacuum freeze-drying method, which was donated as CoFe-AO. As displayed in Fig. 1a, there are obvious peaks at around 16.1,

32.1, 38.4, 39.4, 44.8, and 65.1° for the $\text{CoFe}_n(\text{OH})_x$, which should be ascribed to a bimetallic CoFe-oxyhydroxide. However, it should be noted that these peaks are very weak, indicating low degree of crystallinity. As a contrast, no obvious diffraction peaks can be observed for the CoFe-AO (Fig. S1), indicating its amorphous state. The BET specific area for the $\text{Fe}(\text{OH})_x$, $\text{Co}(\text{OH})_x$, and $\text{CoFe}_{0.8}(\text{OH})_x$ is 390.1 , 43.9 , and $99.2 \text{ m}^2 \text{ g}^{-1}$, respectively (Fig. 1b). From Fig. 1c, irregular blocky structure is typically observed for the $\text{CoFe}_{0.8}(\text{OH})_x$. From HRTEM image (Fig. 1d), partially crystallized structure can be observed at the edges, which is in good agreement with the XRD results. The partially crystallized structure in the $\text{CoFe}_{0.8}(\text{OH})_x$ can be further verified by SAED pattern (Fig. S2). From SEM-EDS mappings (Fig. 2), the Fe and Co elements are well dispersed in the $\text{CoFe}_{0.8}(\text{OH})_x$. Moreover, porous structure can be observed for $\text{Fe}(\text{OH})_x$, $\text{Co}(\text{OH})_x$, and $\text{CoFe}_{0.8}(\text{OH})_x$ (Fig. 2 and Fig. S3).

The elemental composition and the oxidation state of the $\text{CoFe}_{0.8}(\text{OH})_x$ were analyzed by XPS. XPS survey spectrum of $\text{CoFe}_{0.8}(\text{OH})_x$ confirms the existence of Fe, Co, O, C, and Cl elements (Fig. 3a). Combined with the aforementioned SEM-EDX results, it can be inferred that the C element comes from

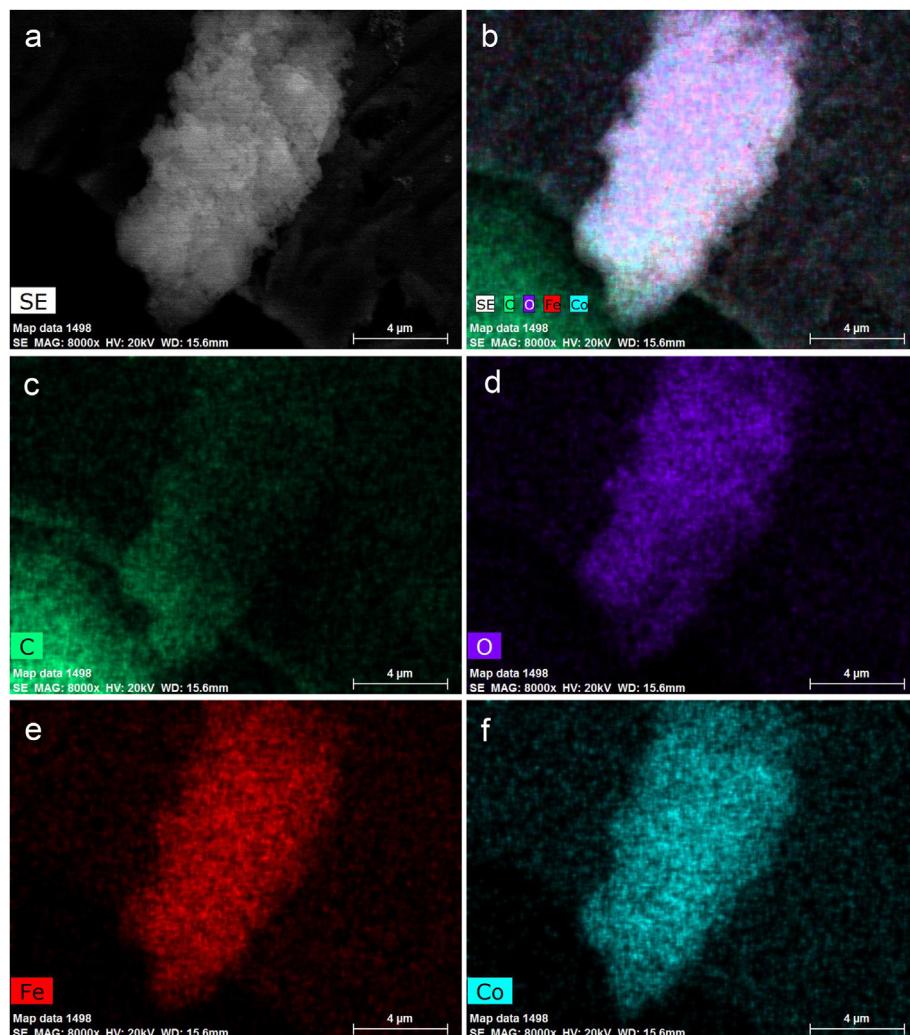


Fig. 2 – (a) SEM image of $\text{CoFe}_{0.8}(\text{OH})_x$. (b–f) EDX mappings.

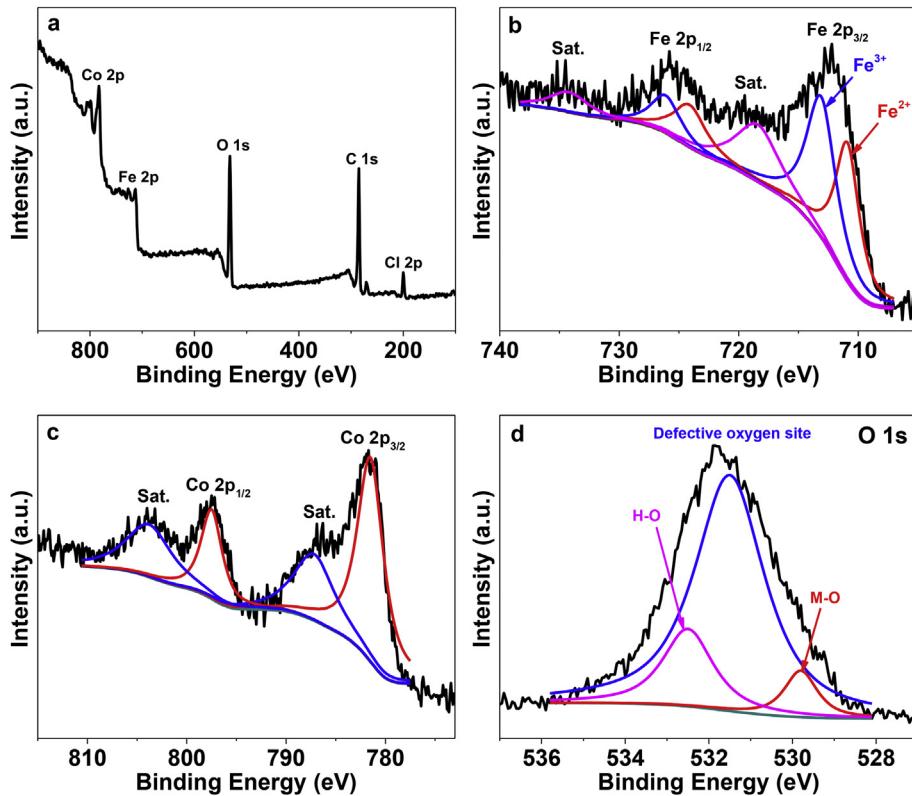


Fig. 3 – (a) XPS survey spectrum of $\text{CoFe}_{0.8}(\text{OH})_x$. **(b-d)** High-resolution Fe 2p, Co 2p, and O 1s XPS spectra, respectively.

polluting carbon sources in the XPS analysis, while the Cl element was absorbed on the catalyst during the synthetic procedure. For the Fe element (Fig. 3b), the Fe 2p_{3/2} peak can be split into Fe^{2+} 2p_{3/2} (710.9 eV) and Fe^{3+} 2p_{3/2} (713.1 eV) peaks

[50]. The relative surface content of $\text{Fe}^{3+}/\text{Fe}^{2+}$ is estimated to be ca. 1.3:1. For the Co element (Fig. 3c), two peaks are located at 781.6 (Co^{2+} 2p_{3/2}) and 797.6 eV (Co^{2+} 2p_{1/2}) along with two satellite peaks [51]. For the O element (Fig. 3d), the O 1s peak can be

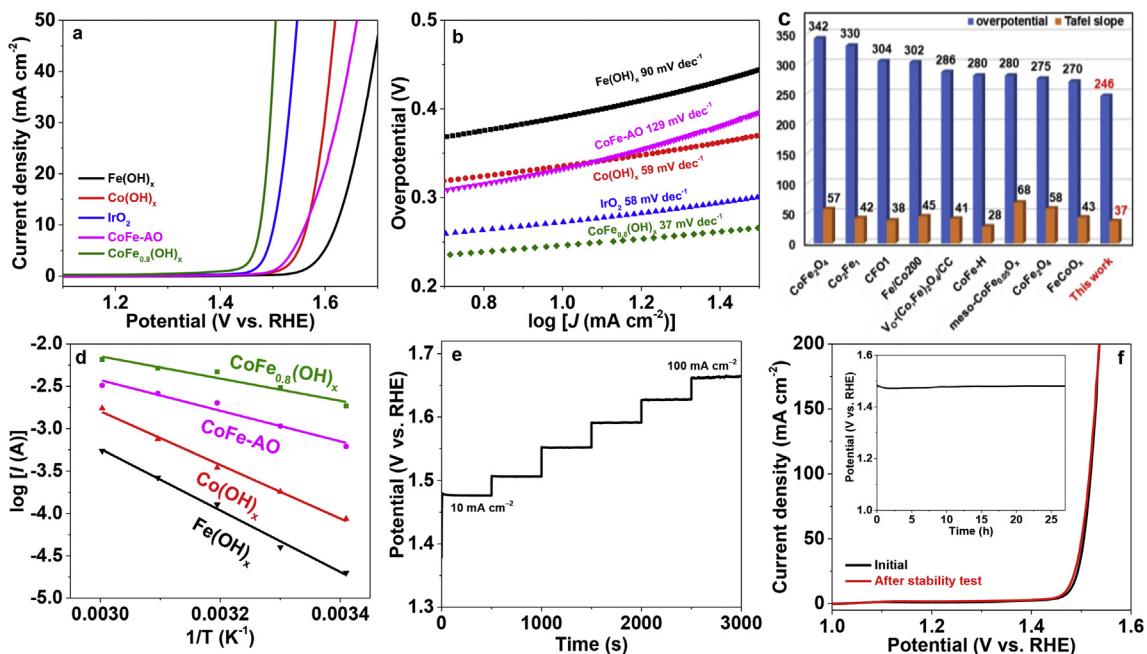


Fig. 4 – (a) Polarization curves. **(b)** Tafel slopes. **(c)** Comparisons of overpotentials at 10 mA cm^{-2} and Tafel slopes. **(d)** Arrhenius plots of the kinetic current at $\eta = 300 \text{ mV}$ without iR compensation. **(e)** Multi-current electrochemical process of $\text{CoFe}_{0.8}(\text{OH})_x$. **(f)** Polarization curves of $\text{CoFe}_{0.8}(\text{OH})_x$ before and after stability test. Inset: Chronopotentiometry curve of $\text{CoFe}_{0.8}(\text{OH})_x$ at a current density of 10 A cm^{-2} .

fitted into three well-defined peaks centered at 532.5 ($-\text{OH}$), 531.5 (defective O site), and 529.8 eV (Fe/Co–O) [28,45,52,53].

The electrocatalytic OER activity of $\text{CoFe}_{0.8}(\text{OH})_x$ was investigated in 1.0 M KOH using a standard three-electrode system with glassy carbon electrode as the working electrode. For comparison, $\text{Fe}(\text{OH})_x$, $\text{Co}(\text{OH})_x$, amorphous $\text{CoFe}_{0.8}$ oxyhydroxide (CoFe-AO), and IrO_2 were also tested under similar conditions. Among them, $\text{CoFe}_{0.8}(\text{OH})_x$ exhibited the best OER performance with the smallest overpotential (η_{10}) of 246 mV at 10 mA cm⁻² and the largest electrocatalytic current through the whole potential range (Fig. 4a), which is lower than the η_{10} of $\text{Fe}(\text{OH})_x$ (391 mV), $\text{Co}(\text{OH})_x$ (335 mV), CoFe-AO (332 mV), and IrO_2 (272 mV), indicating the essential effect of concerted catalysis between Fe and Co and their microenvironment. We further investigated the influence of the molar ratio of Fe/Co on the OER performance. The OER activity increases gradually with increasing the Fe/Co molar ratio and reaches a maximum at $\text{Fe}/\text{Co} = 0.8:1$ (Fig. S4). Further increasing the Fe/Co molar ratio would result in the decrease of OER performance. The Tafel slope of $\text{CoFe}_{0.8}(\text{OH})_x$ is measured as 37 mV dec⁻¹ (Fig. 4b), which is smaller than that of $\text{Fe}(\text{OH})_x$ (90 mV), $\text{Co}(\text{OH})_x$ (59 mV), CoFe-AO (129 mV), and IrO_2 (58 mV), indicating more favorable kinetics for the OER. The η_{10} on the $\text{CoFe}_{0.8}(\text{OH})_x$ is lower than most CoFe-based electrocatalysts for OER reported (Fig. 4c and Table S1), for instance, defect-rich $\text{V}_0\text{-}(\text{Co}, \text{Fe})_3\text{O}_4/\text{CC}$ ($\eta_{10} = 286$ mV) [54], meso-CoFe_{0.05}O_x ($\eta_{10} = 280$ mV) [46], porous CFO1 nanoplates ($\eta_{10} = 304$ mV) [55], Co_2Fe_1 ($\eta_{10} = 330$ mV) [56], Fe/Co200 ($\eta_{10} = 302$ mV) [57], mesoporous CoFe_2O_4 ($\eta_{10} = 342$ mV) [58], CoFe₂O₄ nanospheres ($\eta_{10} = 275$ mV) [59], and amorphous CoFe-H ($\eta_{10} = 280$ mV) [60].

To evaluate the OER barrier on these catalysts, we investigated the influence of temperature on the electrocatalytic activity (Fig. S5). The OER activity increases with elevating the temperature, suggesting that the chemical rate constant is relative to the temperature. The kinetic barrier can be calculated using the Arrhenius plots at $\eta = 300$ mV (Fig. S5) [42,61]. The $\text{CoFe}_{0.8}(\text{OH})_x$ displays a low kinetic barrier of 25.1 kJ mol⁻¹ (Fig. 4d), much smaller than $\text{Fe}(\text{OH})_x$ (69.5 kJ mol⁻¹), $\text{Co}(\text{OH})_x$ (60.5 kJ mol⁻¹), and CoFe-AO (34.6 kJ mol⁻¹), reflecting more favorable OER kinetics on the $\text{CoFe}_{0.8}(\text{OH})_x$. Since monometallic Fe/Co oxyhydroxides show much higher kinetic barrier than bimetallic CoFe oxyhydroxides, the efficient active site for the OER should be di- μ -oxo bridged Fe–Co sites [29].

The electrocatalytic active sites of the catalysts were estimated from the electrochemical double-layer capacitance (Fig. S6). The $\text{CoFe}_{0.8}(\text{OH})_x$ exhibits a double-layer capacitance (C_{dl}) value of 4.68 mF cm⁻², 10.4-fold higher than CoFe-AO under the same conditions, reflecting much more available active sites in the former. Partial crystallization of CoFe oxyhydroxides favors the lattice contraction and the formation of di- μ -oxo bridged Fe–Co sites, thus increasing active sites. Moreover, the C_{dl} increases with increasing the Fe/Co molar ratio and reaches a maximum at $\text{Fe}/\text{Co} = 0.8$. Further increasing the Fe/Co molar ratio would lead to the decrease of C_{dl} value, suggesting that the $\text{CoFe}_{0.8}(\text{OH})_x$ possesses the most active sites in the $\text{CoFe}_n(\text{OH})_x$ samples. The charge transfer ability of the catalysts was studied by electrochemical impedance spectrum (EIS) test (Fig. S7). The charge transfer resistance for $\text{CoFe}_{0.8}(\text{OH})_x$, CoFe-AO, $\text{Co}(\text{OH})_x$, and $\text{Fe}(\text{OH})_x$ is 56, 0.8, 17.6, and 0.5 Ω cm², respectively. Compared

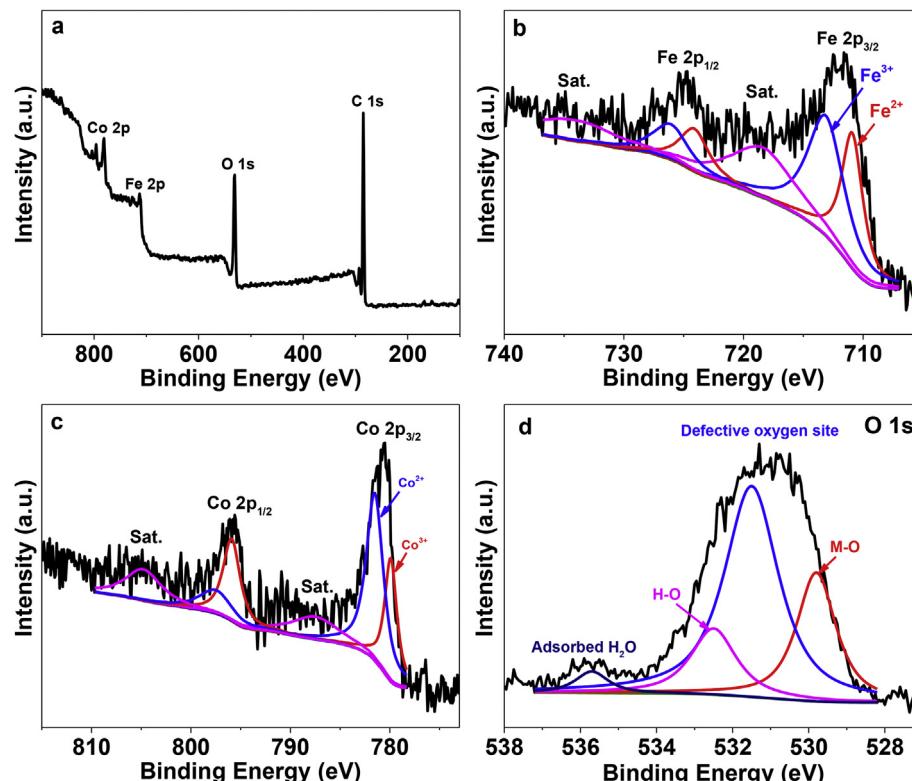


Fig. 5 – XPS survey spectrum (a), and high-resolution XPS spectra of Fe 2p (b), Co 2p (c), and O 1s (d) of the used $\text{CoFe}_{0.8}(\text{OH})_x$.

with amorphous CoFe-AO, partially crystallized $\text{CoFe}_{0.8}(\text{OH})_x$ exhibits much smaller resistance of charge transfer, manifesting more favorable OER kinetics.

The durability of $\text{CoFe}_{0.8}(\text{OH})_x$ was first evaluated multi-step chronopotentiometric measurement (Fig. 4e). The potentials immediately level out at a series of current density ranged from 10 to 100 mA cm^{-2} and remain constant for the residual 500 s, demonstrating that the excellent mass transport ability and robust water oxidation performance of the $\text{CoFe}_{0.8}(\text{OH})_x$. The durability of $\text{CoFe}_{0.8}(\text{OH})_x$ was further assessed by long-time chronopotentiometric measurement at 10 mA cm^{-2} (inset in Fig. 4f), revealing that the $\text{CoFe}_{0.8}(\text{OH})_x$ retains its catalytic activity over 27 h. The polarization curve of the $\text{CoFe}_{0.8}(\text{OH})_x$ after the chronopotentiometric test exhibits no obvious loss (Fig. 4f), implying excellent electrocatalytic OER stability.

The surface compositions and metal valence state of the $\text{CoFe}_{0.8}(\text{OH})_x$ after OER stability measurement were investigated by XPS (Fig. 5). The XPS survey spectrum reveals the existence of Fe, Co, O, and C in the used $\text{CoFe}_{0.8}(\text{OH})_x$ (Fig. 5a). The relative surface content of $\text{Fe}^{3+}/\text{Fe}^{2+}$ increases from 1.3:1 in the fresh sample to 1.5:1 in the used sample (Fig. 5b), indicating that the Fe should be involved in the OER process. Moreover, Co^{3+} ions were generated after OER test and the relative surface content of $\text{Co}^{3+}/\text{Co}^{2+}$ is calculated to be 0.53:1 (Fig. 5c), demonstrating that Co ions also took part in the OER process. Furthermore, the surface oxygen species kept dynamic condition during the OER (Fig. 5d).

Conclusions

In summary, we demonstrate that partially crystallized $\text{CoFe}_{0.8}(\text{OH})_x$ oxyhydroxide exhibited excellent OER activity with an η_{10} of only 246 mV, which is much lower than amorphous CoFe-AO. Kinetic studies revealed that di- μ -oxo bridged Fe-Co sites should be the active sites for efficient OER, over which lower kinetic energy barrier can be achieved than that on single metallic Fe/Co oxyhydroxide. Appropriate crystallization favored the lattice contraction, the generation of Fe-O-Co sites, and the improvement of charge transfer ability, thus promoting OER activity. This work may provide new insights into the design of OER catalysts towards energy-related applications.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.ijhydene.2022.03.174>.

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