

Advanced Optical Materials / Volume 11, Issue 4 / 2202237

Research Article |  Full Access

# An Efficient Perovskite-Like Phosphor with Peak Emission Wavelength at 850 nm for High-Performance NIR LEDs

Wenqiang Tang, Dan Wu , Yu Xiao, Xiaoling Dong, Yue Wang, Wenping Zhou, Yifan Liu, Liangliang Zhang 

First published: 16 December 2022

<https://doi-org.fgul.idm.oclc.org/10.1002/adom.202202237>

Citations: 5

## Abstract

Near-infrared (NIR) phosphor-converted light-emitting diodes (pc-LEDs) are an emerging light source demanded for the miniaturization of NIR spectroscopy. However, the photoelectric efficiency and spectral properties of NIR pc-LEDs are still very limited due to a lack of efficient broadband NIR phosphors with peak emission wavelength >850 nm. Here, a novel Cr<sup>3+</sup>-activated perovskite-like Mg<sub>4</sub>Ta<sub>2</sub>O<sub>9</sub> phosphor is reported, which exhibits a broad NIR emission band peaking at 850 nm with a high internal (external) quantum efficiency of 72% (33.4%). The fabricated NIR pc-LED shows a broad emission band covering the whole 700–1100 nm spectrum range with a photoelectric conversion efficiency of 22.1% at 10 mA. This work may provide a new paradigm for the discovery of Cr<sup>3+</sup> singly doped NIR phosphors with excellent performance.

## 1 Introduction

Near-infrared (NIR) spectroscopy technology is a quantitative analytical method to probe a sample with light in the spectrum range of 700–2500 nm. The mechanism is monitoring fingerprint characteristics absorption of NIR active vibrational modes from heavy atoms such as C, N, O, and S attached to hydrogen.<sup>[1–4]</sup> Considering the spectrum of the most commonly used Si detector for NIR spectroscopy as well as the penetration in bio-tissues, it is important to develop NIR light sources with a wide enough emission spectrum covering the whole 700–1100 nm range to obtain more effective information in practical application such as food analysis, medical diagnostics, and agricultural science.<sup>[5–8]</sup> On the other hand,

integrating the NIR spectroscopy into mobile phones or wearable devices are emerging demand that call for NIR light source with small size.<sup>[9, 10]</sup> The traditional tungsten-halogen lamps or super-continuum lasers cannot meet these new applications because of their large size, short lifetime, and poor efficiency.<sup>[11, 12]</sup> NIR light emitting diodes (LEDs) show the advantages of small size, but their emission bandwidth is less than 50 nm, which is too narrow.<sup>[13, 14]</sup> In this context, phosphor-converted LEDs (pc-LEDs) with broad NIR emission were recently proposed, where one or more kinds of NIR phosphors were coated on a blue LED chip. The broad and tunable emission band as well as the small size of such NIR pc-LEDs make them an excellent alternative light source for miniature NIR spectroscopy applications.

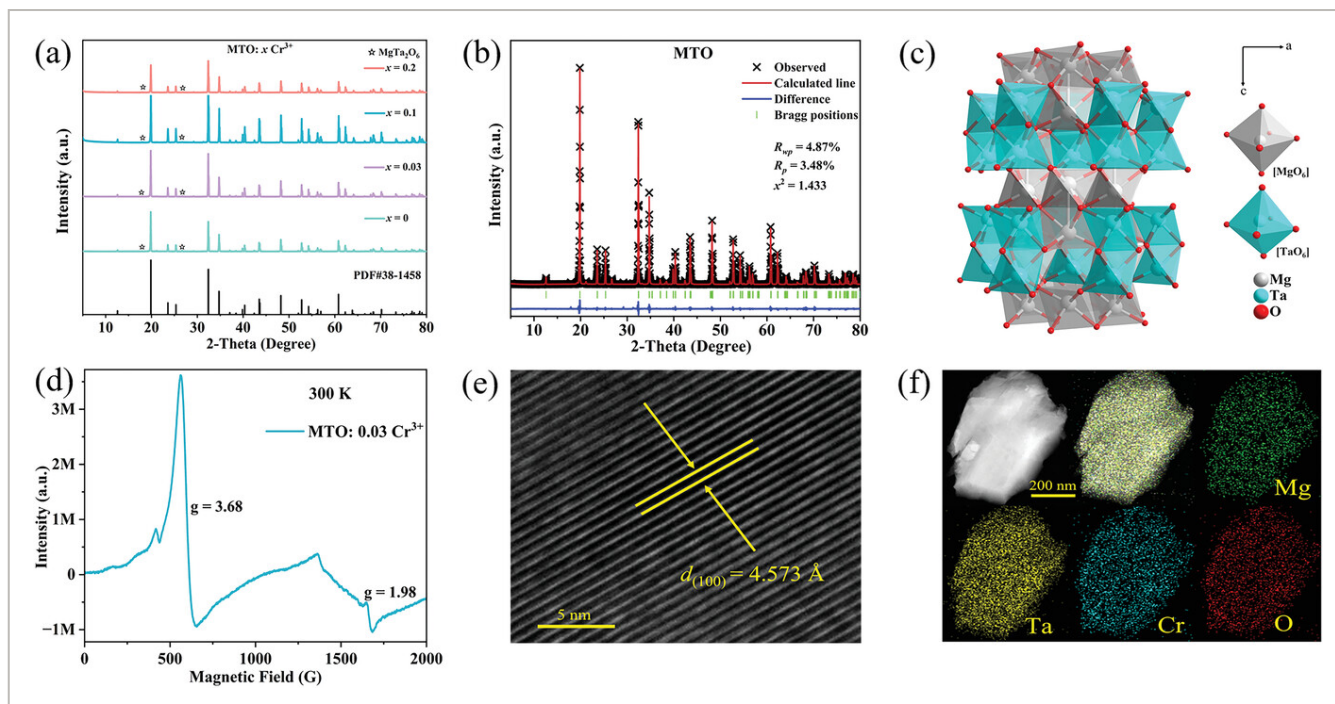
The NIR phosphor acts as a key role to convert blue light from LED chip to broadband NIR light.<sup>[15]</sup> Thus, many efforts have been made to explore NIR phosphors with broad emission band and high efficiency, such as  $\text{K}_3\text{LuSi}_2\text{O}_7\text{:Eu}^{2+}$  (740 nm),<sup>[16]</sup>  $\text{CaS:Eu}^{2+},\text{Tb}^{3+}$  (810 nm)<sup>[17]</sup> and  $\text{Lu}_2\text{BaAl}_4\text{SiO}_{12}\text{:Ce}^{3+},\text{Mn}^{2+}$  (770 nm).<sup>[18]</sup> The main drawbacks of that  $\text{Eu}^{2+}/\text{Ce}^{3+}$  activated phosphor are their relatively short emission wavelength (<850 nm), low internal quantum efficiency (IQE), and poor thermal stability. Thus,  $\text{Cr}^{3+}$ -activated NIR phosphors draw more attention for their broad emission bandwidth and high efficiency.<sup>[19, 20]</sup> For example, Song et al. reported  $\text{K}_2\text{NaScF}_6\text{:Cr}^{3+}$  NIR phosphor with an emission band peaking at 765 nm and IQE of 74%.<sup>[21]</sup> Zheng and co-workers designed  $\text{Y}_2\text{CaAl}_4\text{SiO}_{12}\text{:Cr}^{3+}$  translucent ceramics by cation co-substitution of  $\text{Ca}^{2+}\text{-Si}^{4+}$  for  $\text{Y}^{3+}\text{-Al}^{3+}$  and achieved high IQE (90.1%) and external QE (EQE) (59.5%).<sup>[22]</sup> Highly efficient NIR emissions have also been reported in other phosphors such as  $\text{Ca}_2\text{LuHf}_2\text{Al}_3\text{O}_{12}\text{:Cr}^{3+}$  (775 nm),<sup>[4]</sup>  $\text{La}_2\text{MgZrO}_6\text{:Cr}^{3+}$  (825 nm),<sup>[10]</sup>  $\text{Ga}_2\text{O}_3\text{:Cr}^{3+}$  (740 nm),<sup>[23]</sup>  $\text{GdAl}_3(\text{BO}_3)_4\text{:Cr}^{3+}$  (733 nm),<sup>[24]</sup>  $\text{LaMgGa}_{11}\text{O}_{19}\text{:Cr}^{3+}$  (770 nm)<sup>[25]</sup> and so on. Though the above NIR phosphors have high IQE and excellent thermal stability, they also suffer from deficient emission in 850–1100 nm, which limits their practical applications. Some NIR phosphors with longer wavelength emission (>850 nm), such as  $\text{La}_3\text{Ga}_5\text{GeO}_{14}\text{:Cr}^{3+}$  (920 nm),<sup>[26]</sup>  $\text{ZnTa}_2\text{O}_6\text{:Cr}^{3+}$  (935 nm),<sup>[27]</sup> and  $\text{LiIn}_2\text{SbO}_6\text{:Cr}^{3+}$  (965 nm),<sup>[28]</sup> were reported, but their IQEs are relatively low (typically <30%). Overall, developing broadband NIR phosphors with peak emission wavelength >850 nm and high QE remains a meaningful and challenging target.

In this work, a new broadband perovskite-like NIR phosphor  $\text{Mg}_4\text{Ta}_2\text{O}_9$  (MTO): $\text{Cr}^{3+}$  is reported. The MTO:0.03  $\text{Cr}^{3+}$  phosphor has a broad NIR emission band peaking at 850 nm with full-width at half maximum (FWHM) of 188 nm. The IQE and absorption (Abs) are 72% and 46.4%, respectively, which results in an EQE of 33.4% under 460 nm excitation. The NIR pc-LED fabricated by MTO:0.03  $\text{Cr}^{3+}$  phosphor shows a NIR emission band covering the whole 700–1100 nm spectrum range. The NIR output power is 5.7 mW@10 mA with a photoelectric conversion efficiency of 22.1%.

## 2 Results and Discussion

### 2.1 Structure Analysis

The representative X-ray diffraction (XRD) patterns of  $\text{MTO}:x\text{Cr}^{3+}$  ( $x = 0\text{--}0.06$ ) were monitored and analyzed by the Rietveld refinements to confirm the phase purity and crystal structures (**Figure 1a,b**; Figure S1, Supporting Information). The diffraction peaks can be accurately assigned to  $\text{Mg}_4\text{Ta}_2\text{O}_9$  (PDF#38-1458), though several diffraction peaks of the impurity phase  $\text{MgTa}_2\text{O}_6$  (denoted by a star) also appeared. The comparison of the emission of  $\text{MTO}:0.03\text{Cr}^{3+}$  with that of  $\text{MgTa}_2\text{O}_6:0.03\text{Cr}^{3+}$  as well as the photos of the above two samples are shown in Figure S2 (Supporting Information), and the results indicate that the existent impurity phase has no significant influence on the luminescence properties of  $\text{MTO}:\text{Cr}^{3+}$ . The corresponding refinement results are summarized in Tables S1, S2 (Supporting Information), and the acceptable  $R$ -factors confirm the reliability of the refinements. MTO crystallizes in a hexagonal structure with a space group of (165). According to the above results, the lattice parameters and the cell volumes show a decreasing trend with  $\text{Cr}^{3+}$  concentration because the smaller  $\text{Cr}^{3+}$  ( $r = 0.62\text{ \AA}$ , coordination number (CN) = 6) substitutes the larger  $\text{Mg}^{2+}$  ( $r = 0.72\text{ \AA}$ , CN = 6) or  $\text{Ta}^{5+}$  ( $r = 0.64\text{ \AA}$ , CN = 6), which leads to the lattice shrinkage.<sup>[29-31]</sup> As depicted in Figure 1c, in the unit cell of MTO, two cations of Mg and Ta are in the form of  $(\text{MgO}_6)$  and  $(\text{TaO}_6)$  octahedral by sharing corner oxygens.<sup>[32-37]</sup> Generally, both the octahedral sites of  $\text{Mg}^{2+}$  and  $\text{Ta}^{5+}$  can expect to be occupied by  $\text{Cr}^{3+}$  with  $3d^3$  orbitals due to the similar ionic radius, but considering the different valence states, we designed that  $\text{Cr}^{3+}$  ions only substitute  $\text{Mg}^{2+}$  sites while equal concentration of  $\text{Li}^+$  ions was introduced into MTO to substitute  $\text{Mg}^{2+}$  for the valence balance, additionally, a second low-temperature sintering step under reducing atmosphere was also utilized to avoid  $\text{Cr}^{3+}$  being oxidized into  $\text{Cr}^{6+}$ .<sup>[16]</sup> The electron paramagnetic resonance (EPR) spectrum was also given to probe the local site symmetry of  $\text{Cr}^{3+}$ . As shown in Figure 1d, the EPR spectrum of  $\text{MTO}:0.03\text{Cr}^{3+}$  contains two kinds of signals at  $g$  values of 3.68 and 1.98, respectively. The signal at  $g = 3.68$  is attributed to the isolated  $\text{Cr}^{3+}$  ions, and several signals are observed for the isolated  $\text{Cr}^{3+}$  ions, indicating the existence of several types of  $\text{Cr}^{3+}$  sites. The weak but broad signal at  $g = 1.98$  is assigned to the exchange coupled  $\text{Cr}^{3+}$  pairs, which reflects a disturbance environment of  $\text{Cr}^{3+}\text{--}\text{Cr}^{3+}$  pairs.<sup>[10, 38]</sup> Moreover, according to the refinement results, the average isotropic displacement parameters length of the Mg—O bond dramatically decreases with  $\text{Cr}^{3+}$  content, but that of the Ta—O bond keeps almost unchanged, demonstrating that the  $\text{Cr}^{3+}$  ions mainly enter into the  $\text{Mg}^{2+}$  site. Similar results have been proved in other materials.<sup>[32, 39]</sup>



**Figure 1**

[Open in figure viewer](#) | [Download PowerPoint](#)

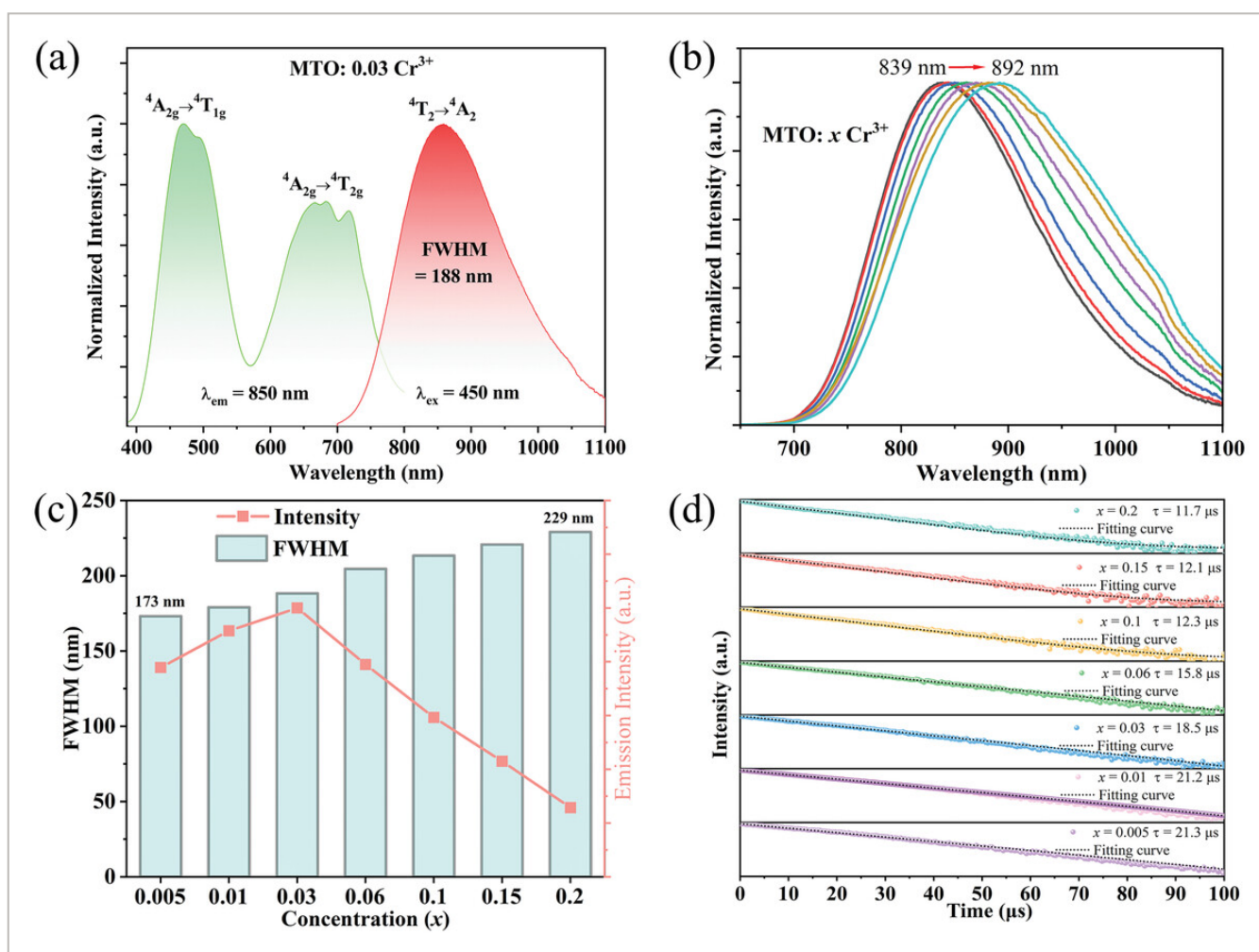
a) XRD patterns of MTO: $x$   $\text{Cr}^{3+}$  ( $x = 0-0.2$ ); b) Rietveld refinement XRD of MTO; The crystal structure and the coordination environments (c), EPR spectrum (d), HRTEM images (e) as well as morphology and mapping (f) of MTO:0.03  $\text{Cr}^{3+}$ .

As shown in Figure 1e, a large number of lattice fringes can be clearly observed in a high-resolution transmission electron microscopy (HRTEM) image, indicating the high crystallinity of MTO:0.03  $\text{Cr}^{3+}$ . The d-spacing value of the (100) plane is measured as 4.573 Å, which is consistent well with its theoretical value of 4.470 Å. An energy dispersive spectrometer (EDS) was utilized to analyze the chemical composition and the relevant results are shown in Figure S3 and Table S3 (Supporting Information). In Figure 1f, the elemental mapping results depict that Mg, Ta, Cr, and O were uniformly distributed throughout the MTO:0.03  $\text{Cr}^{3+}$  sample.

## 2.2 Photoluminescence Properties

**Figure 2a** depicts the excitation and emission spectra of MTO:0.03  $\text{Cr}^{3+}$  at room temperature. Under 450 nm excitation, MTO:0.03  $\text{Cr}^{3+}$  exhibits a single broadband NIR emission peaking at 850 nm with an FWHM of 188 nm ( $2502 \text{ cm}^{-1}$ ), attributing to spin-allowed d-d ( ${}^4\text{T}_2 \rightarrow {}^4\text{A}_2$ ) transition of  $\text{Cr}^{3+}$  in a weak octahedral environment.<sup>[40]</sup> The corresponding excitation spectrum ( $\lambda_{\text{em}} = 850 \text{ nm}$ ) includes two excitation bands at 460 and 680 nm, originating from  ${}^4\text{A}_2 \rightarrow {}^4\text{T}_1$  and  ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$  transitions of  $\text{Cr}^{3+}$ , respectively.<sup>[41]</sup> The blue excitation band as well as the NIR emission band indicate the potential for NIR pc-LED. Additionally, the Tanabe–Sugano diagram is given in Figure S4a (Supporting Information) to

clearly understand the crystal field strength of octahedrally coordinated  $\text{Cr}^{3+}$  in the MTO host. The crystal field strength and Racah parameters,  $Dq$  and  $B$ , were calculated and the results are shown in Supporting Information. The ratio of  $Dq/B$  is calculated to be 1.91, confirming that  $\text{Cr}^{3+}$  ions are in a weak crystal field site with a  ${}^4\text{T}_2$  state lowering than that of  ${}^2\text{E}$ , and thus only the broadband emission ascribed to the  ${}^4\text{T}_2 \rightarrow {}^4\text{A}_2$  transition of  $\text{Cr}^{3+}$  can be detected. The weak crystal field of  $\text{Cr}^{3+}$  in MTO can also be demonstrated by the only dominated broad  ${}^4\text{T}_2 \rightarrow {}^4\text{A}_2$  band at 77 K (Figure S4b, Supporting Information). To optimize the proper doping site of  $\text{Cr}^{3+}$ , the comparison of the emission spectra between  $\text{Cr}^{3+}$  ions only substitute  $\text{Mg}^{2+}$  and  $\text{Cr}^{3+}$  ions simultaneously substitute  $\text{Mg}^{2+}$  and  $\text{Ta}^{5+}$  ( $\text{Mg}_{3.98}\text{Ta}_{1.99}\text{O}_9$ ; 0.03  $\text{Cr}^{3+}$ ) are given in Figure S5 (Supporting Information), the higher emission intensity further indicates that  $\text{Cr}^{3+}$  ions prefer to substitute  $\text{Mg}^{2+}$  ions.



**Figure 2**

[Open in figure viewer](#) | [PowerPoint](#)

a) The room temperature excitation and emission spectra of MTO:0.03 Cr<sup>3+</sup>; b) The normalized emission spectra and c) the FWHM and emission intensities of MTO:x Cr<sup>3+</sup> under 450 nm excitation; d) Fluorescence decay curves of Cr<sup>3+</sup> in MTO:x Cr<sup>3+</sup> (excited at 460 nm, monitored at 850 nm).



Various Cr<sup>3+</sup> doped MTO phosphors were synthesized to demonstrate the luminescence properties and the corresponding emission spectra were also recorded. As depicted in Figure 2b, under 450 nm excitation, the peak position shifts from 839 nm ( $x = 0.005$ ) to 892 nm ( $x = 0.2$ ) while the corresponding FWHM increases from 173 nm ( $2394\text{ cm}^{-1}$ ) to 229 nm ( $2807\text{ cm}^{-1}$ ), which may cause by the multiple octahedral environments of Cr<sup>3+</sup> in MTO and the energy transfer among Cr<sup>3+</sup> ions at high doping level. In Figure 2c, the emission intensity reaches the maximum when  $x$  is 0.03, after that it decreases gradually due to the well-known concentration quenching effect.<sup>[42]</sup> The corresponding excitation spectra as well as the diffuse reflection (DR) spectra were also shown in Figure S6a,b (Supporting Information), and the obvious reflection bands correspond well with the found in the excitation spectrum. The room temperature fluorescence decay curves of MTO: $x$  Cr<sup>3+</sup> monitored at 850 nm are shown in Figure 2d. The decay curves can be well fitted by the single exponential profile, and the lifetime of Cr<sup>3+</sup> monotonically decreases with the increase of the concentration, which also demonstrates the energy transfer among Cr<sup>3+</sup> ions.<sup>[43]</sup>

The IQE and EQE are key parameters directly determining the possibility of phosphors for LED application. The QE of MTO:0.03 Cr<sup>3+</sup> was measured and the corresponding spectrum was shown in Figure S7 (Supporting Information). The IQE value of MTO:0.03 Cr<sup>3+</sup> under 460 nm excitation was measured to be 72% while the Abs is 46.4%, resulting in an EQE value of 33.4%. The Abs of MTO:0.03 Cr<sup>3+</sup> with perovskite-like structure is higher than that of most Cr<sup>3+</sup>-activated phosphor with garnet structure, and its EQE is also higher than most of the multi-site Cr<sup>3+</sup>-activated systems with peak emission wavelength >850 nm (see Table 1 for detailed comparison).

**Table 1.**  $\lambda_{\text{em}}$ , FWHM, IQE, Abs, EQE, NIR output power and photoelectric efficiency of MTO:0.03 Cr<sup>3+</sup> and some reported phosphors

| Samples                                                                                     | $\lambda_{\text{em}}$<br>[nm] | FWHM<br>[nm] | IQE<br>[%] | Abs<br>[%] | EQE<br>[%] | NIR output power | Photoelec<br>efficien |
|---------------------------------------------------------------------------------------------|-------------------------------|--------------|------------|------------|------------|------------------|-----------------------|
| MTO:0.03 Cr <sup>3+</sup>                                                                   | 850                           | 188          | 72         | 46.4       | 33.4       | 60.9 mW@150 mA   | 22.1%@100 mA          |
| GaTaO <sub>4</sub> :0.006 Cr <sup>3+</sup>                                                  | 840                           | 140          | 91.2       | 34.3       | 31.3       | 178 mW@500 mA    | 6%@500 mA             |
| LiInSi <sub>2</sub> O <sub>6</sub> :6% Cr <sup>3+</sup>                                     | 840                           | 143          | 75         | \          | \          | 51.6 mW@100 mA   | 17.8%@100 mA          |
| LiGaP <sub>2</sub> O <sub>7</sub> :12% Cr <sup>3+</sup>                                     | 846                           | 170          | 47.8       | 59.2       | 28.3       | 28.1 mW@120 mA   | 7.8%@120 mA           |
| LaSc <sub>2.4</sub> Ga <sub>0.6</sub> B <sub>4</sub> O <sub>12</sub> :0.07 Cr <sup>3+</sup> | 850                           | \            | 52.1       | \          | \          | \                | \                     |

| Samples                                                              | $\lambda_{em}$<br>[nm] | FWHM<br>[nm] | IQE<br>[%] | Abs<br>[%] | EQE<br>[%] | NIR output power   | Photoelec<br>efficien |
|----------------------------------------------------------------------|------------------------|--------------|------------|------------|------------|--------------------|-----------------------|
| $Sr_9Ga(PO_4)_7:0.8\ Cr^{3+}$                                        | 850                    | \            | 66.3       | 45.1       | 29.9       | 19.79 mW@150<br>mA | 12.3%<br>mA           |
| $(Mg_{0.6}Li_{0.4})$<br>$(Mg_{0.6}Sc_{0.4})Ge_2O_6:4\%$<br>$Cr^{3+}$ | 874                    | 182          | 66.6       | 47.3       | 31.5       | \                  | \                     |

### 2.3 Photoluminescence Thermal Stability

Since the generated heat of LED chips and stokes loss of phosphors will decrease the emission intensity, thermal quenching is another key assessment index for phosphors. To give insight into the thermal quenching behavior of  $MTO:0.03\ Cr^{3+}$ , the temperature-dependent emission spectra were further investigated in **Figure 3**. The emission intensity of  $MTO:0.03\ Cr^{3+}$  experiences a continuous reduction, and still keeps 58.4% of its initial value as the temperature is elevated from 30 °C to 100 °C. Additionally, the FWHM of the NIR emission increases from 188 nm ( $2502\ cm^{-1}$ ) at 30 °C to 240 nm ( $3323\ cm^{-1}$ ) at 300 °C, which may attribute to the increased electron-phonon interaction.<sup>[58]</sup> To clearly understand the thermal quenching behavior, the activation energy  $\Delta E_a$  of  $MTO:0.03\ Cr^{3+}$  was calculated based on the Arrhenius formula and given in Figure S8 (Supporting Information).  $\Delta E_a$  is determined to be 0.345 eV, which is larger than most  $Cr^{3+}$ -activated longer wavelength emission NIR phosphors (Table 1), indicating the good thermal stability of  $MTO:0.03\ Cr^{3+}$ .

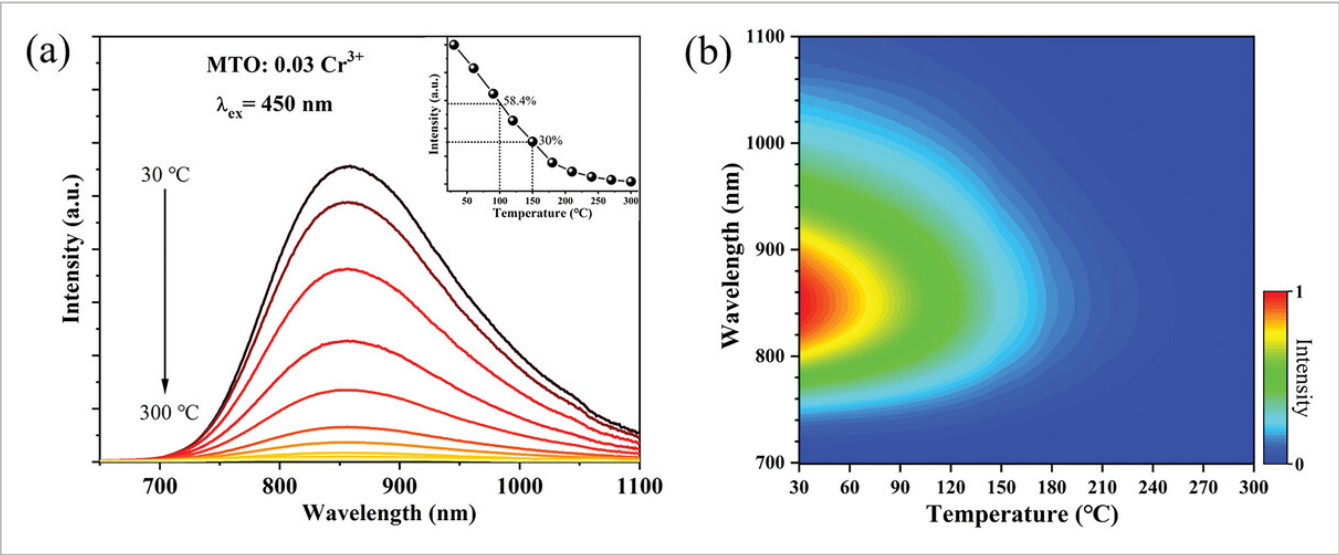


Figure 3

Open in figure viewer

PowerPoint

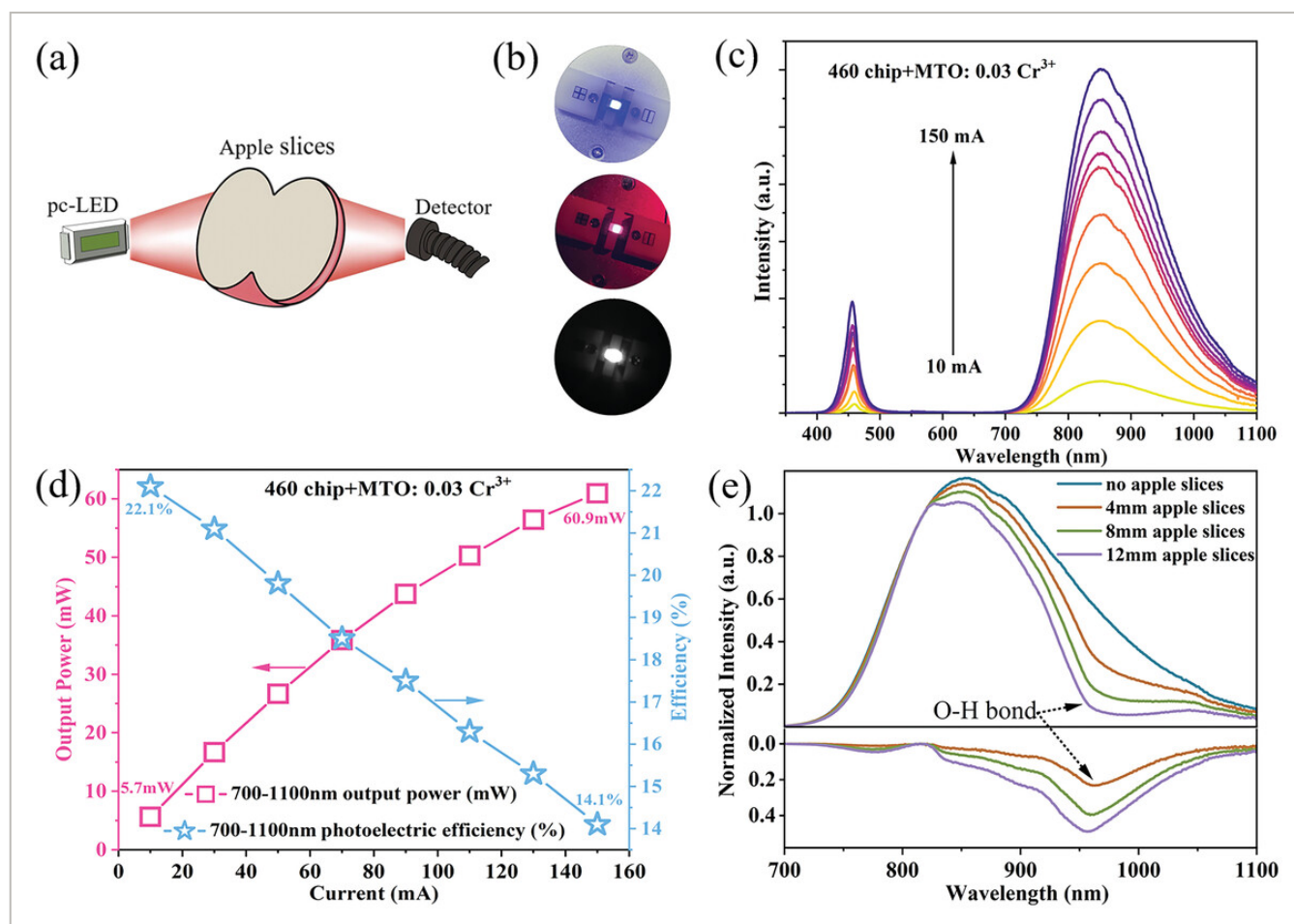
a) Temperature-dependent emission spectra and emission intensities of  $MTO:0.03\ Cr^{3+}$  recorded upon excitation at 450 nm in the temperature range from 30 °C to 300 °C; b) The corresponding contour plot of the emission spectra (y-

axis) and temperature (x-axis) for MTO:0.03 Cr<sup>3+</sup>.

## 2.4 Photoelectric Performance and Applications of NIR LEDs

To prove the application potential of MTO:Cr<sup>3+</sup>, a NIR pc-LED was fabricated by combining a 460 nm LED chip with the optimized MTO:0.03 Cr<sup>3+</sup>, and a home-made testing system was also designed, as simplified in **Figure 4a**. **Figure 4b** shows the photographs of the lightened device. The electroluminescence spectrum contains a broadband NIR emission peaking at 850 nm and a narrow blue emission at 460 nm (**Figure 4c**). The driven current-dependent NIR light output power and photoelectric efficiency of the NIR pc-LED device are also given in **Figure 4d**. The NIR light output power increases from 5.7 mW (10 mA) to 60.9 mW (150 mA). Additionally, the NIR photoelectric efficiencies are calculated to be 22.1% and 14.1% at 10 and 150 mA, respectively, which are higher than that of the most reported NIR pc-LEDs with peak emission wavelength >850 nm (see **Table 1**). The “efficiency droop” of the blue LED chip mainly results in the above decrease in photoelectric efficiency (see **Figure S9**, Supporting Information).<sup>[59, 60]</sup> **Figure 4e** shows the normalized emission spectra of the fabricated NIR pc-LED (0.5 W) via the light penetrating the thin apple slices. The absorption in the range of 900–1050 nm is assigned to the O-H chemical group of the apple, and it can be obtained by the comparison of the measured emission spectra with and without apple slices.<sup>[61]</sup> The emission intensity decreases with the increase of the thickness of the apple slices, and when the thickness is 12 mm, the detector can also clearly detect the signal. The superior performance of the LED device indicates the application potential of MTO:Cr<sup>3+</sup> as an NIR light source.





**Figure 4**

[Open in figure viewer](#) | [PowerPoint](#)

a) Schematic diagram of experimental equipment; b) Photographs of lightened pc-LED without and with a long pass filter at 650 nm; c) Electroluminescence spectrum of the device under the current ranges from 10 to 150 mA; d) Driven current dependent NIR light output power and photoelectric efficiency of the fabricated NIR pc-LED; e) The normalized emission spectra obtained by irradiating the apple slices with the fabricated NIR pc-LED.

### 3 Conclusion

In summary, a series of MTO: $x$  Cr<sup>3+</sup> ( $x = 0.005$ – $0.2$ ) phosphors were successfully synthesized by high-temperature solid-state reactions. As a result of both synthesis strategy and the unique crystal structure of the host lattice MTO, a high IQE and Abs of 72% and 46.4% were achieved. MTO:Cr<sup>3+</sup> performs a broadband NIR emission in the whole 700–1100 nm region with the peak located at 850 nm and FWHM of 188 nm. In addition, the NIR pc-LEDs prepared using the developed phosphors have excellent optical properties, high NIR output power (5.7 mW@10 mA), and high photoelectric efficiency (22.1%@10 mA). Our results provide a promising NIR phosphor for NIR light sources with broad commercial application prospects.

## 4 Experimental Section

### Synthesis

A series of samples of MTO: $x$  Cr<sup>3+</sup> ( $x = 0.005$ – $0.2$ ) were prepared by high-temperature solid-state reaction. MgO (99.99%), Ta<sub>2</sub>O<sub>5</sub> (99.99%), and Cr<sub>2</sub>O<sub>3</sub> (99.999%) were selected as raw materials and weighted in stoichiometric ratios. Li<sup>+</sup> was also used as the charge compensation for the substitution of Mg<sup>2+</sup> by Cr<sup>3+</sup>, and 100% excess of Li<sub>2</sub>CO<sub>3</sub> was added for the evaporation loss during the sintered process. The raw materials were ground in an agate mortar and then transferred into a corundum crucible. The resulting mixtures were sintered in air at 1550 °C for 6 h and subsequently sintered under a reducing atmosphere (95% N<sub>2</sub>-5% H<sub>2</sub>) at 1400 °C for 4 hours to obtain the final powders. The NIR pc-LED was fabricated by encapsulating the prepared MTO:0.03 Cr<sup>3+</sup> onto a blue LED chip (460 nm, 0.5 W, SMD 5730 PCB), where the mass ratio of MTO:0.03 Cr<sup>3+</sup> to the transparent silicone was fixed at 1:1.

### Characterization

XRD patterns were determined by a Rigaku diffractometer (Ultima-IV, with Cu K $\alpha$  radiation) in the  $2\theta$  range from 5° to 120°. The Rietveld refinement analysis was performed on the general structure analysis system program by using Mg<sub>4</sub>Ta<sub>2</sub>O<sub>9</sub> (PDF#38-1458) as a starting model. The HRTEM (JEM-2100F, JEOL) with an EDS was used to characterize the microstructure and the compositions of the samples. The room temperature excitation and emission spectra, temperature-dependent emission spectra as well as the photoelectric properties of the fabricated NIR pc-LEDs device were given by LHS-1000 (EVERFINE) equipped with an array spectrophotometer (HAAS-2000), where the temperature was controlled by a temperature controller (TAP-02, Orient KOJI). The room temperature emission intensities were obtained by using HITACHI F-7000. The excitation and emission spectra at 77 K were measured by FLS980 (Edinburgh Instruments). A UV–VIS–NIR spectrometer (UV3600, Shimadzu) was used to obtain the DR spectra. The EPR measurement at room temperature was performed by Bruker (A300). The fluorescence decays of Cr<sup>3+</sup> were determined by an FS5 fluorimeter (Edinburgh Instruments) (the excitation source was a 450 nm laser diode). The IQE and absorption were recorded by a quantum yield measurement system (Quantaaurus-QY Plus C13534-12, Hamamatsu Photonics).

## Acknowledgements

This work was partially supported by the National Natural Science Foundation of China (Grant No. 11904186; 12104231), the Research Program of Science and Technology at Universities of Inner Mongolia (Grant No. NJZZ20001), the Natural Science Foundation of Jiangsu Province (BK20200783).

## Conflict of Interest

The authors declare no conflict of interest.

Open Research

Data Availability Statement

The data that support the findings of this study are available from the corresponding author upon reasonable request.

Supporting Information

| Filename                                                    | Description            |
|-------------------------------------------------------------|------------------------|
| <a href="#">adom202202237-sup-0001-SuppMat.pdf</a> 899.3 KB | Supporting information |

Please note: The publisher is not responsible for the content or functionality of any supporting information supplied by the authors. Any queries (other than missing content) should be directed to the corresponding author for the article.

References

1

C. Wu, Y. Wu, X. Zhu, J. Zhang, J. Liu, Y. Zhang, *Nano Today* 2021, **36**, 100963.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

2

Y. Liu, S. He, D. Wu, X. Dong, W. Zhou, *ACS Appl. Electron. Mater.* 2022, **4**, 643.

[CAS](#) | [Google Scholar](#)

3

H. Cai, S. Liu, Z. Song, Q. Liu, *J. Mater. Chem. C* 2021, **9**, 5469.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

4

J. Wu, W. Zhuang, R. Liu, Y. Liu, T. Gao, C. Yan, M. Cao, J. Tian, X. Chen, *J. Rare Earth* 2021, **39**, 269.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

5 D. Huang, H. Zhu, Z. Deng, H. Yang, J. Hu, S. Liang, D. Chen, E. Ma, W. Guo, *J. Mater. Chem. C* 2021, **9**, 164.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

6 J. Zhong, Y. Zhuo, F. Du, H. Zhang, W. Zhao, J. Brgoch, *ACS Appl. Mater. Interfaces* 2021, **13**, 31835.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

7 H. Xu, G. Bai, K. He, S. Tao, Z. Lu, Y. Zhang, S. Xu, *Mater. Today Phys.* 2022, **28**, 100872.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

8 L. Zhou, Z. Lyu, D. Sun, S. Shen, T. Tan, L. Wang, H. Zhao, H. You, *Adv. Opt. Mater.* 2022, **10**, 2201308.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

9 R. Y. Li, Y. F. Liu, C. X. Yuan, G. Leniec, L. J. Miao, P. Sun, Z. H. Liu, Z. H. Luo, R. Dong, J. Jiang, *Adv. Opt. Mater.* 2021, **9**, 2100388.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

10 H. Zeng, T. Zhou, L. Wang, R.-J. Xie, *Chem. Mater.* 2019, **31**, 5245.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

11 H. Zhang, J. Zhong, C. Li, L. Wang, W. Zhao, *J. Lumin.* 2022, **251**, 119211.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

12 S. Miao, Y. Liang, Y. Zhang, D. Chen, X. J. Wang, *ACS Appl. Mater. Interfaces* 2021, **13**, 36011.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

13 L. Zhang, D. Wang, Z. Hao, X. Zhang, G. h. Pan, H. Wu, J. Zhang, *Adv. Opt. Mater.* 2019, **7**, 1900185.

[Web of Science®](#) | [Google Scholar](#)

14 H. Zhang, J. Zhong, F. Du, L. Chen, X. Zhang, Z. Mu, W. Zhao, *ACS Appl. Mater. Interfaces* 2022, **14**, 11663.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

15 Q. Zhang, G. Li, P. Dang, D. Liu, D. Huang, H. Lian, J. Lin, *J. Mater. Chem. C* 2021, **9**, 4815.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

16 J. Qiao, G. Zhou, Y. Zhou, Q. Zhang, Z. Xia, *Nat. Commun.* 2019, **10**, 5267.

[PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

17 J. J. Joos, D. Van der Heggen, L. Martin, L. Amidani, P. F. Smet, Z. Barandiaran, L. Seijo, *Nat. Commun.* 2020, **11**, 3647.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

18 Y. Xiao, W. Xiao, D. Wu, L. Guan, M. Luo, L. D. Sun, *Adv. Funct. Mater.* 2022, **32**, 2109618.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

19 Z. Wu, X. Han, J. Wang, Y. Zhou, K. Xing, S. Cao, J. Zhao, B. Zou, R. Zeng, *J. Mater. Chem. C* 2022, **10**, 10292.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

20 Q. Zhang, D. Liu, P. Dang, H. Lian, G. Li, J. Lin, *Laser Photonics Rev.* 2022, **16**, 2100459.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

21 E. Song, H. Ming, Y. Zhou, F. He, J. Wu, Z. Xia, Q. Zhang, *Laser Photonics Rev.* 2021, **15**, 2000410.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

22 G. Zheng, W. Xiao, J. Wu, X. Liu, H. Masai, J. Qiu, *Adv. Sci.* 2022, **9**, 2105713.

[CAS](#) | [Google Scholar](#)

23 M.-H. Fang, G. N. A. De Guzman, Z. Bao, N. Majewska, S. Mahlik, M. Grinberg, G. Leniec, S. M. Kaczmarek, C.-W. Yang, K.-M. Lu, H.-S. Sheu, S.-F. Hu, R.-S. Liu, *J. Mater. Chem. C* 2020, **8**, 11013.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)



24 B. Malysa, A. Meijerink, T. Jüstel, *J. Lumin.* 2016, **171**, 246.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

25 S. Liu, Z. Wang, H. Cai, Z. Song, Q. Liu, *Inorg. Chem. Front.* 2020, **7**, 1467.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

26 V. Rajendran, M.-H. Fang, G. N. D. Guzman, T. Lesniewski, S. Mahlik, M. Grinberg, G. Leniec, S. M. Kaczmarek, Y.-S. Lin, K.-M. Lu, C.-M. Lin, H. Chang, S.-F. Hu, R.-S. Liu, *ACS Energy Lett.* 2018, **3**, 2679.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

27 S. He, P. Li, Y. Ren, G. Wei, Y. Wang, Y. Yang, R. Li, J. Li, Y. Shi, X. Shi, Z. Wang, *Inorg. Chem.* 2022, **61**, 11284.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

28 D. Liu, G. Li, P. Dang, Q. Zhang, Y. Wei, H. Lian, M. Shang, C. C. Lin, J. Lin, *Angew. Chem., Int. Ed.* 2021, **60**, 14644.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

29 Y. Liu, J. Gao, W. Shi, X. Feng, Z. Zhou, J. Wang, J. Guo, R. Kang, B. Deng, R. Yu, *Ceram. Int.* 2021, **47**, 18814.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

30 H. Cai, H. Chen, H. Zhou, J. Zhao, Z. Song, Q. L. Liu, *Mater. Today Chem.* 2021, **22**, 100555.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

31 D. Liu, G. Li, P. Dang, Q. Zhang, Y. Wei, L. Qiu, M. S. Molokeev, H. Lian, M. Shang, J. Lin, *Light: Sci. Appl.* 2022, **11**, 112.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

32 G. Liu, M. S. Molokeev, B. Lei, Z. Xia, *J. Mater. Chem. C* 2020, **8**, 9322.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

33 H. Suo, Y. Wang, X. Zhao, X. Zhang, L. Li, K. Guan, W. Ding, P. Li, Z. Wang, F. Wang, *Laser Photonics Rev.* 2022, **16**, 2200012.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

34 L. Lou, S. Zhao, S. Yuan, D. Zhu, F. Wu, Z. Mu, *Inorg. Chem. Front.* 2022, **9**, 3522.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

35 R. E. Newnham, J. F. Dorrian, E. P. Meagher, *Mater. Res. Bull.* 1970, **5**, 199.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

36 A. J. Brown, J. Liu, F. P. Marlton, M. Avdeev, B. J. Kennedy, C. D. Ling, *J. Solid State Chem.* 2020, **287**, 121385.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

37 R. Mani, N. S. P. Bhuvanesh, K. V. Ramanujachary, W. Green, S. E. Lofland, J. Gopalakrishnan, *J. Mater. Chem.* 2007, **17**, 1589.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

38 Z. Jia, C. Yuan, Y. Liu, X. J. Wang, P. Sun, L. Wang, H. Jiang, J. Jiang, *Light: Sci. Appl.* 2020, **9**, 86.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

39 M. Mao, T. Zhou, H. Zeng, L. Wang, F. Huang, X. Tang, R.-J. Xie, *J. Mater. Chem. C* 2020, **8**, 1981.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

40 E. T. Basore, W. Xiao, X. Liu, J. Wu, J. Qiu, *Adv. Opt. Mater.* 2020, **8**, 2000296.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

41 L. Zhang, S. Zhang, Z. Hao, X. Zhang, G.-h. Pan, Y. Luo, H. Wu, J. Zhang, *J. Mater. Chem. C* 2018, **6**, 4967.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

42 D. Wu, H. Wu, Y. Xiao, X. Dong, Y. Wang, W. Zhou, Y. Liu, L. Zhang, *J. Lumin.* 2022, **244**, 118750.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

43 Y. Zhang, S. Miao, Y. Liang, C. Liang, D. Chen, X. Shan, K. Sun, X. J. Wang, *Light: Sci. Appl.* 2022, **11**, 136.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

44 J. Zhong, Y. Zhuo, F. Du, H. Zhang, W. Zhao, S. You, J. Brgoch, *Adv. Opt. Mater.* 2022, **10**, 2101800.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

45 X. Xu, Q. Shao, L. Yao, Y. Dong, J. Jiang, *Chem. Eng. J.* 2020, **383**, 123108.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

46 C. Yuan, R. Li, Y. Liu, L. Zhang, J. Zhang, G. Leniec, P. Sun, Z. Liu, Z. Luo, R. Dong, J. Jiang, *Laser Photonics Rev.* 2021, **15**, 2100227.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

47 T. Gao, W. Zhuang, R. Liu, Y. Liu, C. Yan, X. Chen, *Cryst. Growth Des.* 2020, **20**, 3851.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

48 F. Zhao, H. Cai, Z. Song, Q. Liu, *Chem. Mater.* 2021, **33**, 3621.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

49 H. Liu, F. Zhao, H. Cai, Z. Song, Q. Liu, *J. Mater. Chem. C* 2022, **10**, 9232.

[Web of Science®](#) | [Google Scholar](#)

50 L. Yao, Q. Shao, S. Han, C. Liang, J. He, J. Jian, *Chem. Mater.* 2020, **32**, 2430.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

51 T. Liu, H. Cai, N. Mao, Z. Song, Q. Liu, *J. Am. Ceram. Soc.* 2021, **104**, 4577.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

52 M. Zhao, S. Liu, H. Cai, F. Zhao, Z. Song, Q. Liu, *Sci. China Mater.* 2022, **65**, 748.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

53 X. Zhou, W. Geng, J. Li, Y. Wang, J. Ding, Y. Wang, *Adv. Opt. Mater.* 2020, **8**, 1902003.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

54 W. Zhou, J. Luo, J. Fan, H. Pan, S. Zeng, L. Zhou, Q. Pang, X. Zhang, *Ceram. Int.* 2021, **47**, 25343.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

55 D. Dai, Z. Wang, Z. Xing, X. Li, C. Liu, L. Zhang, Z. Yang, P. Li, *J. Alloys Compd.* 2019, **806**, 926.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

56 D. Wen, H. Liu, Y. Guo, Q. Zeng, M. Wu, R. S. Liu, *Angew. Chem., Int. Ed.* 2022, **61**, e202204411.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

57 F. Zhao, Z. Song, J. Zhao, Q. Liu, *Inorg. Chem. Front.* 2019, **6**, 3621.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

58 T. Senden, R. J. A. van Dijk-Moes, A. Meijerink, *Light: Sci. Appl.* 2018, **7**, 8.

[PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

59 G. Zheng, W. Xiao, H. Wu, J. Wu, X. Liu, J. Qiu, *Laser Photonics Rev.* 2021, **15**, 2100060.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

60 F. He, E. Song, H. Chang, Y. Zhou, Z. Xia, Q. Zhang, *ACS Appl. Mater. Interfaces* 2022, **14**, 31035.

[CAS](#) | [PubMed](#) | [Web of Science®](#) | [Google Scholar](#)

61 T. Ma, X. Li, T. Inagaki, H. Yang, S. Tsuchikawa, *J. Food Eng.* 2018, **224**, 53.

[CAS](#) | [Web of Science®](#) | [Google Scholar](#)

Citing Literature



[Download PDF](#)

ABOUT WILEY ONLINE LIBRARY

- Privacy Policy
- Terms of Use
- About Cookies
- Manage Cookies
- Accessibility
- Wiley Research DE&I Statement and Publishing Policies
- Developing World Access

HELP & SUPPORT

- Contact Us
- Training and Support
- DMCA & Reporting Piracy

OPPORTUNITIES

- Subscription Agents
- Advertisers & Corporate Partners

CONNECT WITH WILEY

- The Wiley Network
- Wiley Press Room

Copyright © 1999-2024 John Wiley & Sons, Inc or related companies. All rights reserved, including rights for text and data mining and training of artificial technologies or similar technologies.